

Passivation of Highly Boron Doped Silicon Surfaces by Sputtered AlO_x and PECVD SiN_x , a Comparison

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Abstract: We show that boron-diffused emitters can be passivated with AlO_x deposited using RF sputtering of an Al target. The surface passivation achieved so far is inferior to that obtained using an optimised PECVD SiN_x process that includes a chemically grown SiO_2 interfacial layer. Nevertheless, the levels of passivation obtained, expressed by emitter recombination current densities of $J_{0E}=228\text{--}349 \text{ fA/cm}^2$ for sheet resistances of $88\text{--}210 \text{ }\Omega/\square$, are already consistent with solar cell with efficiencies in the 20% range.

1 Introduction: An emerging application of aluminium oxide, AlO_x , is the passivation of the front, boron diffused surface of n-type silicon solar cells. Atomic Layer Deposition techniques have given the best results so far [1,2], thanks to a low density of interface states combined with a high negative charge. In this experiment, we investigate the passivation of boron-diffused surfaces using AlO_x deposited by RF sputtering of an Al target [3] and compare it to the passivation achieved with an optimised PECVD SiN_x process [4].

2 Experiment: 4", $10 \text{ }\Omega\cdot\text{cm}$, (100) Cz n-type silicon wafers were saw-damage etched using a 50% TMAH solution and chemically polished using a 10:1 HNO_3/HF solution to a final thicknesses of $370\text{--}470 \mu\text{m}$. The boron diffusions were performed in a quartz tube furnace using BBr_3 at 865°C or at 900°C for 60 min. To obtain a wider range of sheet resistances and surface dopant concentrations, some of the wafers were annealed at 1000°C in N_2 for 3 h or 18 h, after having removed the boron glass using alternating steps in hot HNO_3 and HF solutions. The boron diffusions with no drive-in step give an "industrial type" emitter, with a high sur-

face doping and a relatively shallow pn junction. Those that underwent a drive-in step give a more optimised, "laboratory type", dopant profile with relatively low surface doping and deep junction.

The 4" wafers were quartered to provide a set of samples for passivation with sputtered AlO_x and another set for passivation with PECVD SiN_x . The latter was performed at the Energy research Centre of the Netherlands (ECN) using a method reported by Mihailetchi et al. [4] that includes the chemical growth of a thin SiO_2 layer using nitric acid prior to the deposition of the SiN_x . We will refer to this as a NAOS/ SiN_x stack. AlO_x was deposited at the ANU using a 99.999% pure Al target, reactively sputtered with 2 sccm of O_2 and 20 sccm of Ar at a pressure of 3 mT. The RF power was 298 W, with intrinsic bias voltages of around 120 V.

The samples with sputtered AlO_x were annealed in a quartz tube furnace at 425°C in N_2 for 30 min. Those having a SiN_x passivation were "fired" in a conveyor belt furnace at approximately 800°C . The dielectric layers were deposited on both sides of the wafers to create a symmetrical structure in order to extract the values of the emitter recombination current density J_{0E} corresponding to the boron emitter regions using photoconductance characterisation techniques [5].

3 Results and discussion: Fig. 1 shows the measured saturation current density, J_{0E} , corresponding to the p+ emitter region (inclusive of bulk and surface recombination) for different boron diffusions and for two types of surface passivation, either a sputtered AlO_x layer (approximately 20 nm thick) or a nitric acid $\text{SiO}_2/\text{SiN}_x$ stack (NAOS/ SiN_x).

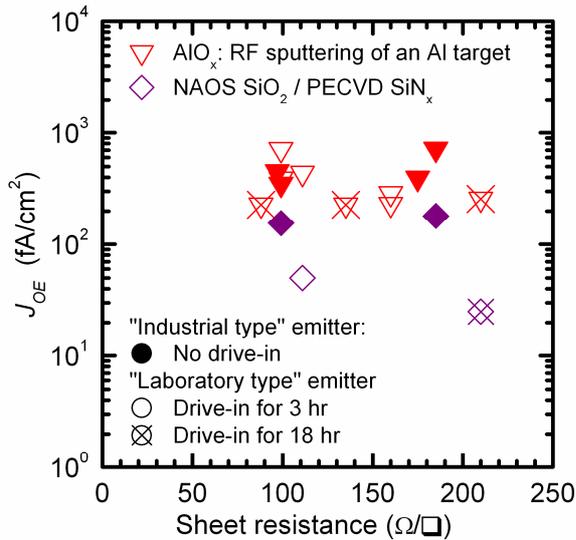


Fig. 1. Recombination (saturation) current density J_{0E} of surface passivated boron-diffused emitters as a function of their sheet resistance.

For the “industrial type” emitter, the best value of J_{0E} using sputtered AlO_x is 349 fA/cm^2 at a sheet resistance of $100 \text{ } \Omega/\square$. This is consistent with achieving open-circuit voltages of up to 655 mV in n-type ($1 \text{ } \Omega\cdot\text{cm}$) solar cells. Its corresponding sister sample that had been passivated with the NAOS/ SiN_x stack achieved a J_{0E} of 155 fA/cm^2 . This indicates that further optimization of the “industrial type” emitter profile is needed, since the values achieved by Mihailetchi et al. [4] for a similar sheet resistance are much lower. The results do, nevertheless, prove that sputtered AlO_x is capable of passivating boron diffused surfaces.

For the “laboratory type” emitters, the best value of J_{0E} using sputtered AlO_x is $228\text{--}257 \text{ fA/cm}^2$ for a range of sheet resistances between $88 \text{ } \Omega/\square$ and $210 \text{ } \Omega/\square$. This is consistent with achieving open-circuit voltages of up to 665 mV in n-type ($1 \text{ } \Omega\cdot\text{cm}$) solar cells. The control samples that had been passivated with the nitric acid $\text{SiO}_2/\text{SiN}_x$ stack achieved a very low value of $J_{0E} = 25 \text{ fA/cm}^2$ at $210 \text{ } \Omega/\square$, indicating that in this case the surface passivation achieved with sputtered AlO_x are not limited by the boron profile itself and that further optimisation of the sputtering process is required.

Table 1: Emitter saturation currents measured for different boron diffusions and passivations.

sample		NAOS /SiN	NAOS/ AlO_x	AlO_x
Diffusion	Resist.	J_{0e}	J_{0e}	J_{0e}
temp (C)	ohm/sq	fA/cm ²	fA/cm ²	fA/cm ²
900/1h	100	155	477	349
900/1h +1000/3h	112	50	873	237
865/1h	184	180	868	716
865/1h +1000/18h	210	25	542	257

3 Conclusions: This work has demonstrated a reasonable level of passivation of various boron-diffused surfaces using AlO_x deposited by RF sputtering of an Al target. J_{0E} values as low as 228 fA/cm^2 across sheet resistances of $88\text{--}210 \text{ } \Omega/\square$ have been achieved in this initial experiment; better surface passivation can be expected with the optimisation of the boron diffusion and AlO_x deposition.

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