

# Passivation of Highly Boron Doped Silicon Surfaces by Sputtered $\text{AlO}_x$ and PECVD $\text{SiN}_x$ , a Comparison

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**Abstract:** We show that boron-diffused emitters can be passivated with  $\text{AlO}_x$  deposited using RF sputtering of an Al target. The surface passivation achieved so far is inferior to that obtained using an optimised PECVD  $\text{SiN}$  process that includes a chemically grown  $\text{SiO}_2$  interfacial layer. Nevertheless, the levels of passivation obtained, expressed by emitter recombination current densities of  $J_{0E}=228\text{--}349\text{ fA/cm}^2$  for sheet resistances of  $88\text{--}210\text{ }\Omega/\square$ , are already consistent with solar cell with efficiencies in the 20% range.

**1 Introduction:** An emerging application of aluminium oxide,  $\text{AlO}_x$ , is the passivation of the front, boron diffused surface of n-type silicon solar cells. Atomic Layer Deposition techniques have given the best results so far [1,2], thanks to a low density of interface states combined with a high negative charge. In this experiment, we investigate the passivation of boron-diffused surfaces using  $\text{AlO}_x$  deposited by RF sputtering of an Al target [3] and compare it to the passivation achieved with an optimised PECVD  $\text{SiN}$  process [4].

**2 Experiment:** 4", 10  $\Omega\cdot\text{cm}$ , (100) Cz n-type silicon wafers were saw-damage etched using a 50% TMAH solution and chemically polished using a 10:1  $\text{HNO}_3/\text{HF}$  solution to a final thicknesses of 370-470  $\mu\text{m}$ . The boron diffusions were performed in a quartz tube furnace using  $\text{BBr}_3$  at 865°C or at 900 °C for 60 min. To obtain a wider range of sheet resistances and surface dopant concentrations, some of the wafers were annealed at 1000 °C in  $\text{N}_2$  for 3 h or 18 h, after having removed the boron glass using alternating steps in hot  $\text{HNO}_3$  and HF solutions. The boron diffusions with no drive-in step give an "industrial type" emitter, with a high sur-

face doping and a relatively shallow pn junction. Those that underwent a drive-in step give a more optimised, "laboratory type", dopant profile with relatively low surface doping and deep junction.

The 4" wafers were quartered to provide a set of samples for passivation with sputtered  $\text{AlO}_x$  and another set for passivation with PECVD  $\text{SiN}_x$ . The latter was performed at the Energy research Centre of the Netherlands (ECN) using a method reported by Mihailetchi et al. [4] that includes the chemical growth of a thin  $\text{SiO}_2$  layer using nitric acid prior to the deposition of the  $\text{SiN}_x$ . We will refer to this as a NAOS/ $\text{SiN}_x$  stack.  $\text{AlO}_x$  was deposited at the ANU using a 99.999% pure Al target, reactively sputtered with 2 sccm of  $\text{O}_2$  and 20 sccm of Ar at a pressure of 3 mT. The RF power was 298 W, with intrinsic bias voltages of around 120 V.

The samples with sputtered  $\text{AlO}_x$  were annealed in a quartz tube furnace at 425 °C in  $\text{N}_2$  for 30 min. Those having a  $\text{SiN}$  passivation were "fired" in a conveyor belt furnace at approximately 800°C. The dielectric layers were deposited on both sides of the wafers to create a symmetrical structure in order to extract the values of the emitter recombination current density  $J_{0E}$  corresponding to the boron emitter regions using photoconductance characterisation techniques [5].

**3 Results and discussion:** Fig. 1 shows the measured saturation current density,  $J_{0E}$ , corresponding to the p+ emitter region (inclusive of bulk and surface recombination) for different boron diffusions and for two types of surface passivation, either a sputtered  $\text{AlO}_x$  layer (approximately 20 nm thick) or a nitric acid  $\text{SiO}_2/\text{SiN}_x$  stack (NAOS/ $\text{SiN}_x$ ).

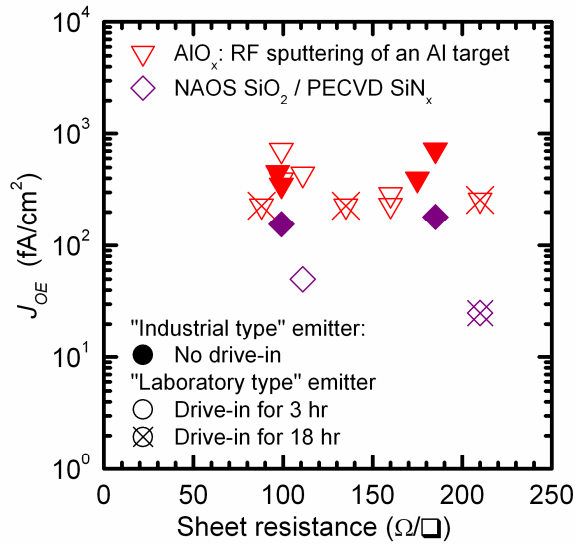


Fig. 1. Recombination (saturation) current density  $J_{0E}$  of surface passivated boron-diffused emitters as a function of their sheet resistance.

For the “industrial type” emitter, the best value of  $J_{0E}$  using sputtered  $\text{AlO}_x$  is  $349 \text{ fA/cm}^2$  at a sheet resistance of  $100 \Omega/\square$ . This is consistent with achieving open-circuit voltages of up to 655 mV in n-type ( $1 \Omega\cdot\text{cm}$ ) solar cells. Its corresponding sister sample that had been passivated with the NAOS/ $\text{SiN}_x$  stack achieved a  $J_{0E}$  of  $155 \text{ fA/cm}^2$ . This indicates that further optimization of the “industrial type” emitter profile is needed, since the values achieved by Mihailetchi et al. [4] for a similar sheet resistance are much lower. The results do, nevertheless, prove that sputtered  $\text{AlO}_x$  is capable of passivating boron diffused surfaces.

For the “laboratory type” emitters, the best value of  $J_{0E}$  using sputtered  $\text{AlO}_x$  is  $228\text{--}257 \text{ fA/cm}^2$  for a range of sheet resistances between  $88 \Omega/\square$  and  $210 \Omega/\square$ . This is consistent with achieving open-circuit voltages of up to 665 mV in n-type ( $1 \Omega\cdot\text{cm}$ ) solar cells. The control samples that had been passivated with the nitric acid  $\text{SiO}_2/\text{SiN}_x$  stack achieved a very low value of  $J_{0E} = 25 \text{ fA/cm}^2$  at  $210 \Omega/\square$ , indicating that in this case the surface passivation achieved with sputtered  $\text{AlO}_x$  are not limited by the boron profile itself and that further optimisation of the sputtering process is required.

Table 1: Emitter saturation currents measured for different boron diffusions and passivations.

sample		NAOS /SiN	NAOS/ $\text{AlO}_x$	$\text{AlO}_x$
Diffusion	Resist.	$J_{0E}$	$J_{0E}$	$J_{0E}$
temp (C)	ohm/sq	fA/cm <sup>2</sup>	fA/cm <sup>2</sup>	fA/cm <sup>2</sup>
900/1h	100	155	477	349
900/1h +1000/3h	112	50	873	237
865/1h	184	180	868	716
865/1h +1000/18h	210	25	542	257

**3 Conclusions:** This work has demonstrated a reasonable level of passivation of various boron-diffused surfaces using  $\text{AlO}_x$  deposited by RF sputtering of an Al target.  $J_{0E}$  values as low as  $228 \text{ fA/cm}^2$  across sheet resistances of  $88\text{--}210 \Omega/\square$  have been achieved in this initial experiment; better surface passivation can be expected with the optimisation of the boron diffusion and  $\text{AlO}_x$  deposition.

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